

Optical, Magnetic and Electrochemical Properties of Spinel Mn_3O_4 , Co_3O_4 Nanostructures

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ABSTRACT

Manganese oxide (Mn_3O_4) and Cobalt oxide (Co_3O_4) nanocrystals were synthesized using a straightforward and cost-effective combustion method. Structural analysis through X-Ray Diffraction, FTIR Spectroscopy, and Raman Spectroscopy confirmed structure to be tetragonal for Mn_3O_4 with space group of $I4_1/amd$ (141) and the cubical for Co_3O_4 with space group $Fd3m$ (227). The usual microscopic crystal sizes were determined using the Debye Scherer formula, resulting in approximately 14 nm for Mn_3O_4 and 22 nm for Co_3O_4 . SEM images displayed irregular and uniform spherical grains for Mn_3O_4 and Co_3O_4 , respectively, with average grain sizes of 100 nm and 120 nm. EDS and XPS spectra showed clear binding energy peaks for Manganese, Cobalt, and Oxygen, confirming the chemical purity of the samples. The optical band gap of the nanocrystal Mn_3O_4 and Co_3O_4 were 2.93 eV, 2.50 eV respectively. Magnetic analysis revealed a strong ferromagnetic nature at 5 K temperature for both Mn_3O_4 and Co_3O_4 nanocrystals. Electrochemical investigations demonstrated that Mn_3O_4 nanocrystals exhibited excellent performance as electrodes for electrochemical capacitors. At a current density of 0.5 A/g, they achieved higher specific capacitance of 417 F/g, maintaining almost 100% coulombic efficiency after 600 cycles and displaying good cyclic stability for up to 10,000 cycles. On the other hand, Co_3O_4 showed inferior electrochemical behavior, indicating that Mn_3O_4 nanocrystals possess superior mass transport of electrolyte ions into the electrode, making them more suitable for electrochemical capacitors.

Keywords—Manganese Oxide & Cobalt Oxide nanocrystals, optical band gap, electrochemical capacitors.

I. INTRODUCTION

Lately, electrochemical capacitors have risen to prominence as potential energy storage solutions, thanks to their favorable attributes including extended cycle longevity and impressive energy and power densities [1]. Prior investigations highlight that the efficacy of supercapacitors is notably shaped by the selection of electrode substance and electrolyte. Transition metal oxides and conductive polymers are frequently harnessed as dynamic electrode materials, yielding substantial capacitance but encountering challenges in sustained cycling efficiency due to faradaic redox reactions [2, 3]. Alternatively, carbonaceous electrodes offer better cyclability but lower capacitance, relying on double-layer capacitance between the electrode and electrolyte.

Among transition metal oxides, some like Ru_2O , MoO_3 , V_2O_5 , Fe_2O_3 , Mn_3O_4 , Co_3O_4 , Cupric Oxide, Zinc Oxide, Manganese Oxide, Nickel oxide, and Titanium dioxide have received extensive attention for their

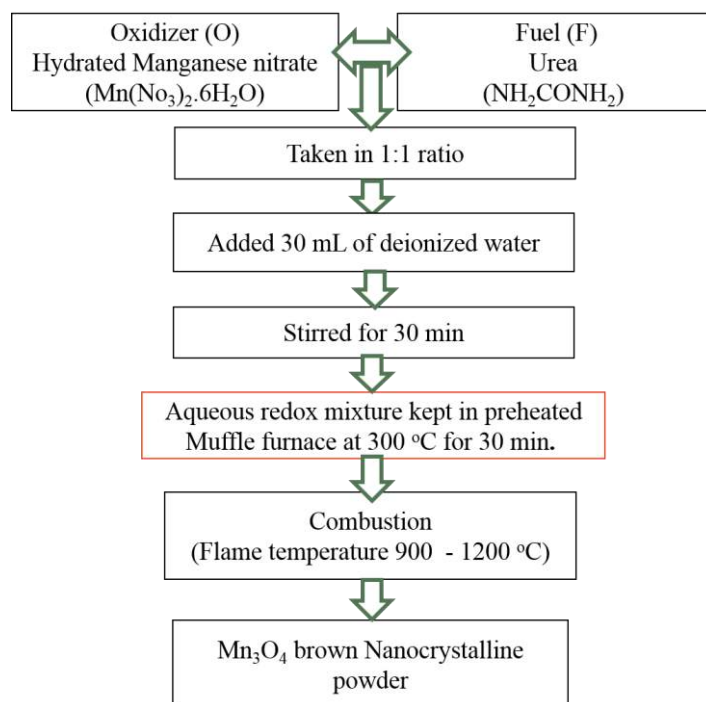
higher specific capacitance theoretically [4, 5]. Among these, RuO_2 is well-studied for its high conductivity, specific capacitance, and chemical stability. However, its limited abundance and high cost hinder practical applications. Spinel ferrites like Mn_3O_4 and Co_3O_4 are considered attractive electrode materials due to their cost-effectiveness, eco-friendliness, and intrinsic high capacity [6-11]. These materials have diversified utility across an array of domains, encompassing rechargeable lithium-ion batteries, molecular adsorption, electrochemistry, magnetism, and supercapacitors. Furthermore, their remarkable catalytic aptitude in oxygen reduction and reaction (ORR) renders them indispensable for ORR catalysis. Beyond that, the distinct magnetic attributes exhibited by these nanoparticles have kindled interest for their conceivable roles in data storage, magnetic particle imaging, wastewater treatment, and other prospective applications [13, 14].

Spinel ferrites and their composites have also found applications in diverse areas, thanks to their unusual magnetodielectric properties. For example, substituted spinel ferrites synthesized through various methods have been utilized in anti-cancer and functional radio electronic devices, catalysis, and biological applications [15-20].

However, there is limited research comparing the magnetic and electrochemical properties of spinel Mn_3O_4 and Co_3O_4 nanocrystals prepared via combustion synthesis. While various methods exist for synthesizing oxide nanoparticles, combustion synthesis stands out due to its simplicity, low temperature, and short processing time. This article aims to investigate and compare the optical, magnetic, ac capacitance, and electrochemical properties of spinel Mn_3O_4 and Co_3O_4 nanocrystals. The study reveals that Mn_3O_4 nanocrystals outperform Co_3O_4 , attributed to their high spin and suitable structure with a smaller crystallite size [23, 24].

II. EXPERIMENTAL PROCEDURE

The reagents utilized in synthesizing spinel Mn_3O_4 nanocrystals were provided by the Merck Company and possessed a purity level of over 99%. The preparation involved using manganese nitrate as the oxidant and urea as the fuel, mixed in a 1:1 molar ratio. The oxidant and fuel mixture was dissolved in 30 ml of deionized water and stirred for approximately 30 minutes using a magnetic stirrer. The resulting solution was collected in a quartz crucible and then placed inside a preheated muffle furnace set at 300 °C for a duration of 30 minutes. After the chemical reaction was complete, the final product was allowed to cool naturally to room temperature and carefully collected. The detailed step-by-step procedure for the solution combustion technique used to prepare Mn_3O_4 nanocrystals is depicted in the accompanying flowchart. The same experimental procedure was replicated for the synthesis of Co_3O_4 nanocrystals.



Schematic diagram of solution combustion technique

A. Materials Characterizations

The structural attributes of the synthesized Mn_3O_4 and Co_3O_4 nanocrystals were investigated employing both X-ray powder diffractometer (XRD-Model 3003 TT) and Raman spectrometer (Horiba Jobin Yvon Lab RAM HR800UV). To deduce the vibrational modes, Fourier Transform Infrared spectrometer (FTIR-ALPHA interferometer, Bruker) was employed. Morphological evaluation along with elemental analysis was executed utilizing Scanning Electron Microscopy (SEM) and Energy Dispersive Spectroscopy (EDS) (Carl ZEISS Model EVO MA15).

For the analysis of chemical states, X-ray Photoelectron Spectrometer (XPS) was employed. Both survey and core-level high-resolution spectra were acquired employing a monochromatic $\text{Al K}\alpha$ X-ray (1486.6 eV) under a vacuum level of 10–9 Torr. The experiment employed an applied beam current of 9 mA and an acceleration voltage of 13 keV (117 W). To investigate the optical band gap, a UV-Vis spectrometer was utilized.

For studying the magnetic properties, Vibrational Sample Magnetometer (VSM) was employed. The electrochemical characteristics of the nanocrystals were evaluated through Cyclic Voltammetry (CV), Chronopotentiometry (CP), and Electrochemical Impedance Spectroscopy (EIS) investigations, which were performed using a 608CHI electrochemical analyzer.

B. Preparation of Electrodes

To construct the electrode, a blend comprising the synthesized Mn_3O_4 , carbon black, and PVdF in an 80:10:10 proportion is subjected to grinding for thirty minutes. The resultant mixture is then integrated with N-

methyl-2-pyrrolidone solvent to yield a slurry. This slurry is applied onto a chemically treated Nickel foam substrate using a brush. The coated electrode is subsequently air-dried at 100°C for a duration of two hours.

To assess the electrochemical performance of the synthesized Mn₃O₄ nanocrystals, a glass cell is constructed. This cell comprises a working electrode embedded with the prepared Mn₃O₄, with a platinum foil serving as the reference electrode, and Ag/AgCl functioning as the counter electrode. The cell is then filled with a 1M Na₂SO₄ aqueous solution. Subsequently, a parallel procedure is executed to prepare the Co₃O₄ nanocrystal electrode, enabling subsequent investigation.

III. RESULTS AND DISCUSSION

A. Microstructural Properties

The structural analysis of the synthesized Mn₃O₄ and Co₃O₄ nanocrystals was conducted through X-ray powder diffraction (XRD) across a Bragg's angle range of 10°-70°. Figure 1a displays the XRD pattern of Mn₃O₄, revealing diffraction peaks at 19.43°, 33.48°, 38.18°, 44.33°, 49.83°, 59.06°, 60.68°, and 64.96°, corresponding to (101), (103), (004), (220), (105), (321), (224), and (440) reflections, respectively. Additionally, a predominant (211) orientation peak was observed at 2θ = 36.3°. These distinct peaks substantiate the tetragonal structure of Mn₃O₄, affiliated with the 141/amd (141) space group as identified by JCPDS card No. 89-4837. The lattice parameters are derived as a = b = 5.763 Å and c = 9.456 Å [15]. The pronounced intensity and broadening of these peaks indicate a notable crystallinity level paired with a relatively smaller crystallite size.

Figure 1b illustrates the XRD pattern of Co₃O₄, which shows a major (311) orientation peak at 2θ = 37.19°, along with other characteristic peaks at 31.54°, 38.68°, 44.59°, 55.31°, 59.55°, and 65.35° corresponding to (220), (222), (400), (422), (333), and (440) reflections. All discernible peaks are aligned with the cubic structure displaying an Fd3m (227) space group, characteristic of a conventional spinel structure. The calculated lattice constant stands at a = b = c = 8.072 Å [15].

The determination of the crystallite size (D) for the samples employed Debye-Scherrer's formula (1), resulting in sizes of 14 nm for Mn₃O₄ and 22 nm for Co₃O₄, correspondingly.

$$D = \frac{k \lambda}{\beta \cos \theta} \quad \text{--- (1)}$$

where k=0.94, λ=1.5406 Å, β=Width of the diffraction peak at half intensity in radian and θ=Bragg's angle in degrees.

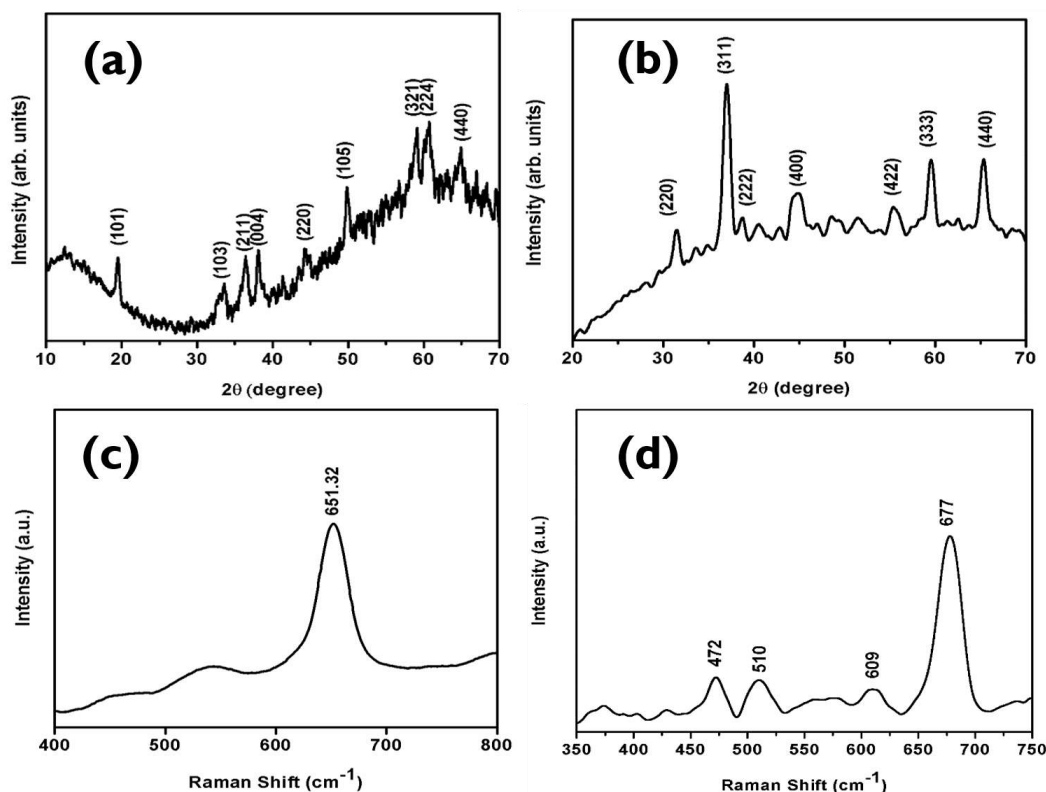


Figure 1: XRD spectra and Raman spectra of spinel (a, c) Mn_3O_4 , (b, d) Co_3O_4

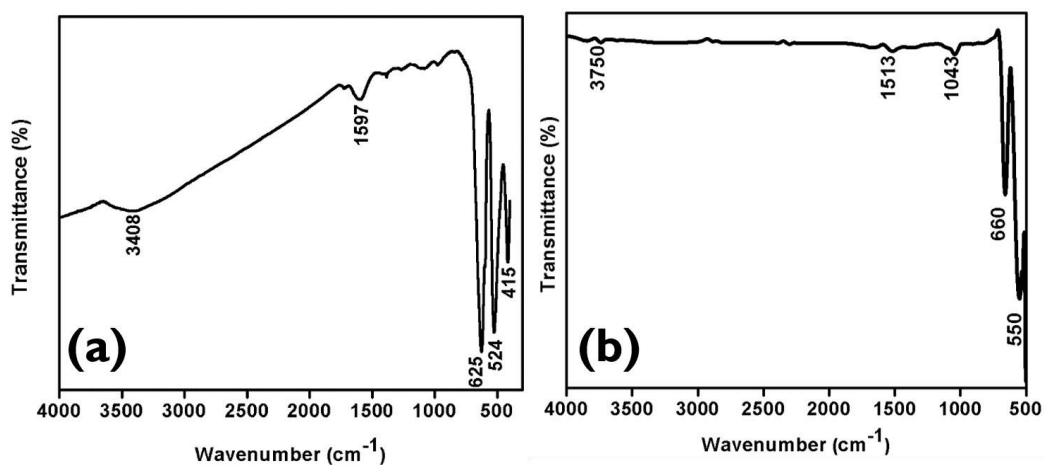


Figure 2 (a): FTIR spectrum of spinel Mn_3O_4 and (b) Co_3O_4

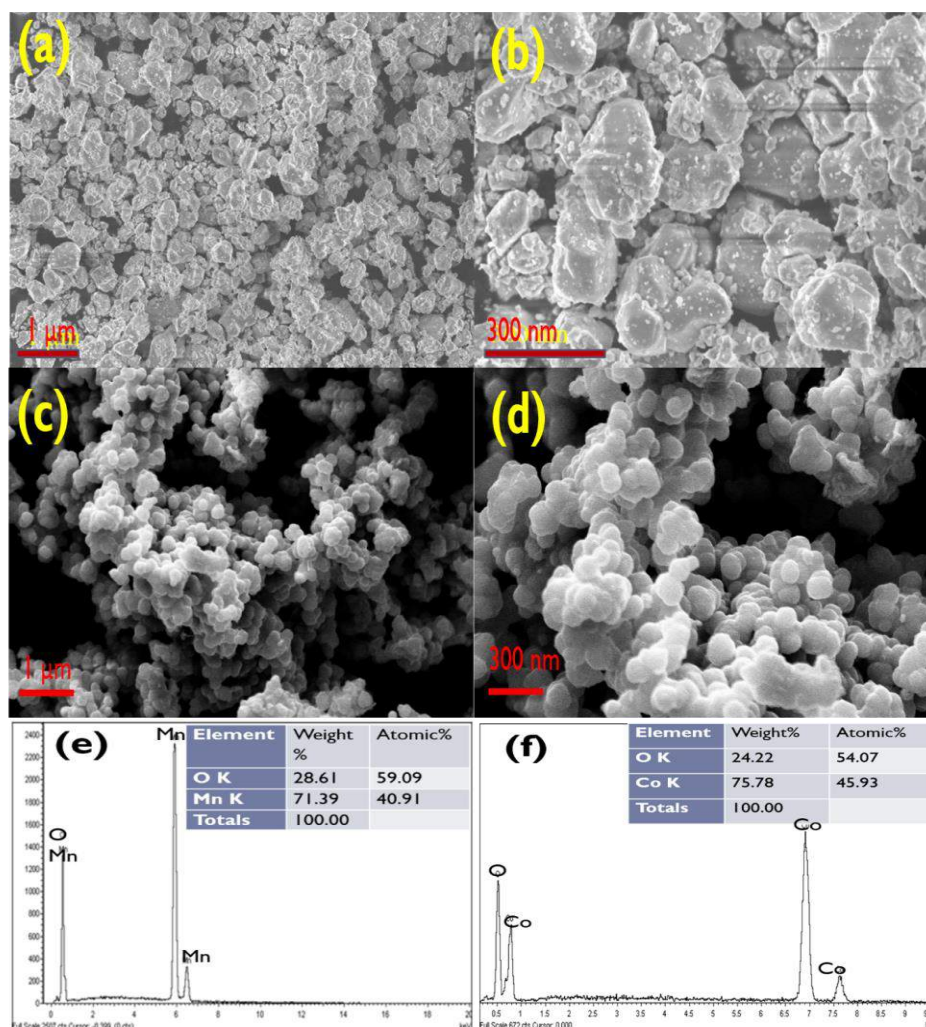
To validate the structure of the synthesized Mn_3O_4 and Co_3O_4 nanocrystals, Raman spectroscopy was implemented. In Figure 1c, the Raman spectrum of Mn_3O_4 unfolds within the wavenumber span of 400-800 cm^{-1} . The discernible peak at 651.32 cm^{-1} corresponds to the singularly degenerate A_{1g} symmetry mode, attributing to the Mn-O stretching vibrations of Mn^{2+} in tetrahedral coordination [25]. A marginal blue shift of this peak signifies a reduced crystallite size, thereby corroborating the findings from XRD [26]. As portrayed in Figure 1(d), the Raman spectrum of Co_3O_4 nanocrystals spans the Raman shift domain of 350-750 cm^{-1} . The robust band near 670 cm^{-1} , along with a cluster of less intense bands spanning 400-600 cm^{-1} , constitutes hallmark features of Co_3O_4 . The potency of the peak at 677 cm^{-1} and the presence of two feeble peaks at 609 and 510 cm^{-1} align with A_{1g} , $F_{2g}(1)$, and $F_{2g}(2)$ vibrations, correspondingly. These vibrations correspond to

Co-O stretching, Co-O-Co asymmetric stretching, and Co-O bending modes. Additionally, a faint resonance at 472 cm⁻¹ signifies E_g, a doubly degenerate symmetry mode. These distinctive peaks validate the cubic structure inherent in Co₃O₄ [27].

FTIR spectra were acquired for the Mn₃O₄ and Co₃O₄ nanocrystals, encompassing wavenumbers ranging from 4000 to 500 cm⁻¹, to assess the existence of vibrational bonds within the samples (Fig. 2a&b). As illustrated in Figure 2a, absorption peaks emerge at 625 and 524 cm⁻¹, correspondingly aligning with the Mn-O stretching mode of tetrahedral and octahedral sites. Furthermore, a feeble absorption peak at 415 cm⁻¹ is linked to the vibration of manganese within an octahedral site. Notably, vibrational bands at 1513 and 3750 cm⁻¹ correspond to O-H bending vibrations [25]. Turning to Figure 2b, absorption peaks at 550 and 660 cm⁻¹ correlate with vibrations of Co³⁺ and Co²⁺ in octahedral and tetrahedral sites, respectively. These FTIR analyses affirm the presence of Mn-O and Co-O bonds, thereby reinforcing the outcomes derived from Raman spectroscopy.

The morphology of the Mn₃O₄ and Co₃O₄ nanocrystals underwent scrutiny via Scanning Electron Microscopy (SEM), with the findings depicted in Figure 3. SEM images reveal the Mn₃O₄ sample's composition as irregularly shaped grains with uneven distribution, featuring a typical size of 100 nm (Fig. 3a&b). In contrast, Co₃O₄ nanocrystals exhibit homogenous distribution, adopting a spherical grain form, with an average sphere size of 120 nm (Figure 3c&d).

For elemental analysis, Energy Dispersive Spectroscopy (EDS) was employed. As depicted in Figure 3e, the binding energy peaks of Manganese and Oxygen present atomic percentages of 40.91 and 59.09, respectively. This equates to a metal-to-oxygen ratio of 0.69 for the Mn₃O₄ sample. Turning to Figure 3f, the binding energy peaks of Cobalt and Oxygen manifest with atomic percentages of 45.93 and 54.09, respectively, resulting in a metal-to-oxygen ratio of 0.85 for the Co₃O₄ sample. These EDS findings substantiate the chemical purity of both the Mn₃O₄ and Co₃O₄ samples.



**Figure 3: (a & b) SEM images of spinel Mn₃O₄ and (c & d) Co₃O₄ nanocrystals
EDS of spinel (e) Mn₃O₄ (f) Co₃O₄ nanocrystals**

X-ray Photoelectron Spectroscopy (XPS) investigations were performed to assess the composition and chemical states of the samples. In the case of the Mn₃O₄ sample, the survey spectrum demonstrates discernible peaks attributed to Mn 3p, Mn 3s, O 1s, and Mn 2p, as illustrated in Figure 4a. The Mn 2p spectrum in Figure 4b displays binding energy peaks at 641.25 eV (Mn 2p_{3/2}) and 653.16 eV (Mn 2p_{1/2}), featuring a separation of 11.91 eV. Figure 4c portrays the Mn 3s spectrum, revealing two low binding energy peaks at 82.62 eV and 88.64 eV, with a gap of 5.71 eV between them. Additionally, the XPS spectrum of O 1s [Figure 4d] manifests around 532 eV, confirming the presence of oxygen within the sample. The observed energy disparities between Mn 3s and Mn 2p peaks harmonize with earlier findings [25].

For the Co₃O₄ sample, Figure 4e captures the Co 2p spectrum, demonstrating peak division at 780 eV (Co 2p_{3/2}) and 795 eV (Co 2p_{1/2}) with a separation of 15 eV [28]. This spectrum solidifies the identification of cobalt and its particular chemical state within the Co₃O₄ nanocrystals.

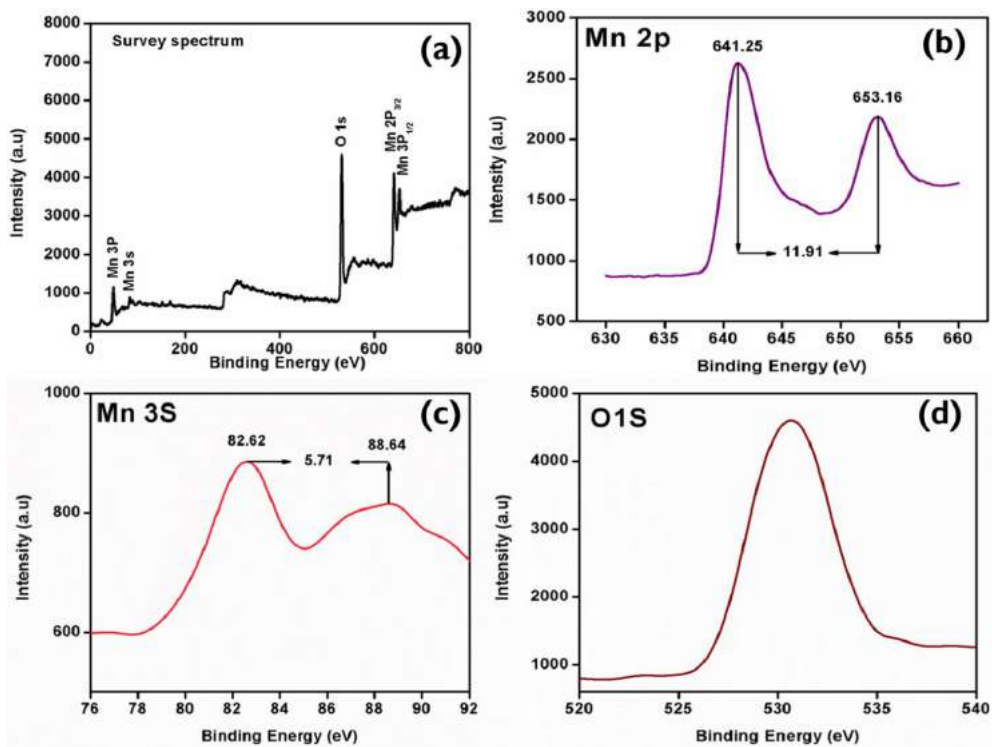


Figure 4: Mn_3O_4 nanocrystals -X-ray Photoelectron spectra (a) Survey spectrum, (b) Mn 2p (c) Mn 3s and (d) O 1s.

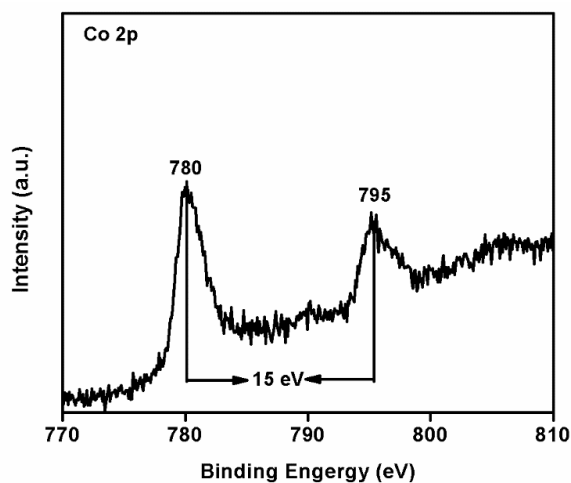


Figure 4(e) XPS spectra of Co_3O_4 nanocrystals

B. Optical and Magnetic properties

UV-Visible spectroscopy was employed to investigate the optical characteristics of the synthesized nanocrystals within the wavelength of about 300-1000 nm to evaluate their optical band gap. The derived spectra for optical band gap, alongside the absorption spectra (visible in the inset of Figure 5), were observed. Demonstrating a semiconductor nature of p-type, the nanocrystals exhibited substantial absorption within the visible range. By utilizing an appropriate equation [29], the optical band gap values for the Mn_3O_4 and Co_3O_4 nanocrystals were calculated to be 2.93 eV and 2.5 eV, respectively.

It's noteworthy that the determined E_g values for Mn_3O_4 and Co_3O_4 nanocrystals in our study are higher than those reported in previous research [30]. This rise in the band gap is attributed to the quantum confinement effects and the presence of oxygen vacancies within the Mn_3O_4 and Co_3O_4 nanoparticles [31-34]. These factors influence electronic structure and energy levels, leading to an elevated optical band gap in the synthesized nanocrystals

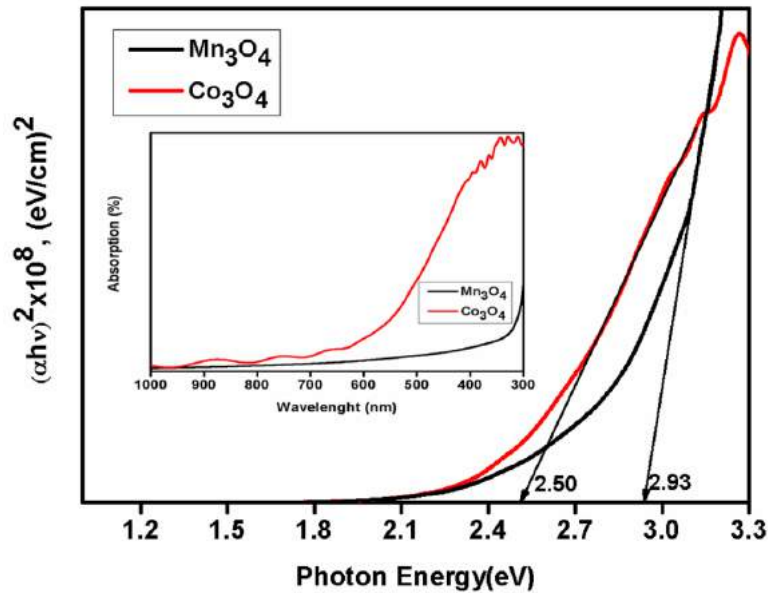


Figure 5: Optical band gaps of Co_3O_4 and Mn_3O_4 nanocrystals

The Vibrating Sample Magnetometer (VSM) was employed to examine the magnetic characteristics of Mn_3O_4 and Co_3O_4 nanocrystals across a temperature span of 5 K to 300 K, as depicted in Figure 6. The magnetic attributes of intricate oxides such as Mn_3O_4 and Co_3O_4 , possessing regular spinel structures, are acknowledged to exhibit significant sensitivity to variations in oxygen content. In our study, the stoichiometry of oxygen of the synthesised Mn_3O_4 and Co_3O_4 nanocrystals was determined to be $Mn_3O_{4.06}$ and $Co_3O_{3.9}$, respectively, indicating oxygen excess and deficit in the respective samples. The stoichiometric $Mn_3O_{4.06}$ sample exhibited ferromagnetic properties, as shown in Fig. 6a&b, with the magnetization curve at 5 K displaying a hysteresis loop characteristic of ferromagnetism. At 300 K, the curve resembled paramagnetic behavior.

The prevailing oxidation state of manganese cations within the octahedral coordination of oxygen cations was determined, predominantly constituting Mn^{3+} (70.9%), accompanied by a lesser proportion of Mn^{4+} (19.1%). In the context of the $Mn_3O_{4.06}$ sample, it was established that the remanence magnetization (M_r) stood at 0.199 emu g^{-1} , while the coercivity (H_c) value was measured at 1.06 kOe.

Conversely, distinct characteristics emerged in the $Co_3O_{3.9}$ sample. Figure 6c & d illustrates its weak ferromagnetic traits, indicated by a saturation magnetization of 0.25 emu g^{-1} under a magnetic field of 0.7 kOe. The transformation from the bulk Co_3O_4 's antiferromagnetic state to the $Co_3O_{3.9}$ nanocrystals' weak ferromagnetic state can be ascribed to surface spins that remain uncompensated or finite size effects. It is well-recognized that the magnetic attributes of nanomaterials hinge greatly on factors such as particle dimensions, shape, crystallinity, and magnetization orientation. Thus, the $Co_3O_{3.9}$ sample can be seen as a ferromagnetic

matrix housing antiferromagnetic clusters, a consequence of the presence of oxygen vacancies that give rise to antiferromagnetic interactions.

In conclusion, the magnetic behavior of nanocrystalline Co_3O_4 and Mn_3O_4 is influenced by oxygen stoichiometry and the presence of oxygen vacancies, leading to weak ferromagnetic and ferromagnetic properties, respectively. The interplay between antiferromagnetic and ferromagnetic interactions leads to the emergence of the spin glass state within the $\text{Co}_3\text{O}_{3.9}$ nanocrystals.

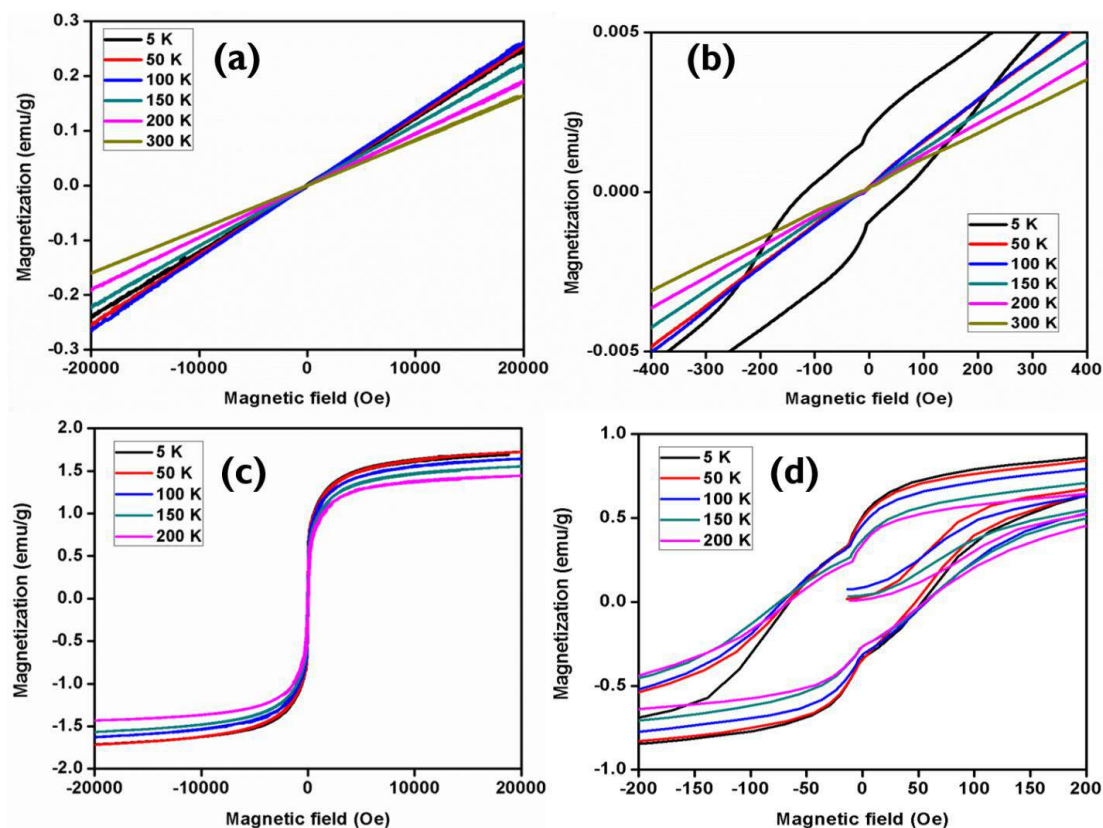


Figure 6- a & b) M–H curves of Mn_3O_4 and (c & d) Co_3O_4 nanocrystals at 5K to 300 K.

C. Dielectric properties

The capacitive response of the $\text{Mn}_3\text{O}_{4.06}$ specimen is examined through an assessment of how capacitance changes with both frequency and potential, as portrayed in Figure 7a and 7b. The graph in Figure 7a illustrates a noticeable pattern: capacitance is more pronounced at lower frequencies but diminishes as frequency escalates. Ultimately, the value stabilizes at higher frequencies. This phenomenon is attributed to the presence of space charge polarization, which loses significance at greater frequencies, thus culminating in a frequency-independent demeanor. Furthermore, the specimen's capacitance registers an upswing with escalating temperatures, as scrutinized across temperatures spanning 313 K to 373 K.

In addition, an exploration into the $\text{Mn}_3\text{O}_{4.06}$ sample's capacitance variance concerning potential at diverse temperatures is depicted in Figure 7b. Evidently, a reduction in oxygen vacancies corresponds to heightened sample conductivity, thereby bestowing enhanced electrochemical characteristics [36].

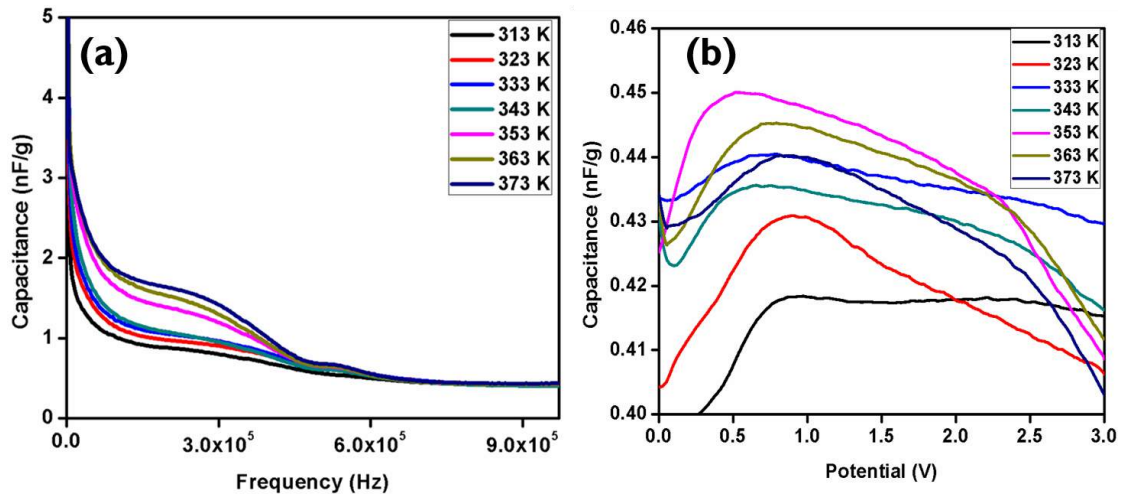


Figure 7: (a) Dependence of AC Capacitance on frequency (b) Capacitance on potential of Mn₃O₄ nanocrystals

D. Electrochemical properties

Nanocrystals of Mn₃O₄ and Co₃O₄, are subjected to Cyclic Voltammetry study in a one Molar Na₂SO₄ aqueous solution. The CV experiments are conducted at different scan rates: 1-50 mV/s for Mn₃O₄ and 1-10 mV/s for Co₃O₄. The potential window range for the experiments is -0.1 V to +0.9 V for Mn₃O₄ and -0.1 V to 0.7 V for Co₃O₄. The cyclic voltammograms obtained from the experiments are described. The CV curves for both Mn₃O₄ and Co₃O₄ show rectangular and semi-rectangular shapes. These shapes suggest that there are no distinct redox peaks in the curves. The absence of redox peaks in the CV curves indicates that the observed behavior might be due to interface capacitance between the electrode and the electrolyte. Interface capacitance is related to the ability of the electrode to store charge at the interface with the electrolyte. When the scan rate increases, the area under the CV curves also increases. This indicates that more charge is being transferred during the experiment. However, the specific capacitance decreases with higher scan rates. This could be because at higher scan rates, the movement of electrolytic ions is limited, resulting in reduced charge storage capacity. Despite the changes in the CV curve characteristics with varying scan rates, the overall shape of the CV curves remains consistent. This consistency suggests that the samples (Mn₃O₄ and Co₃O₄) are chemically stable under the experimental conditions. Equation (2) below refers to a formula used to calculate the specific capacitance (C) of the samples from the cyclic voltammetry (CV) data and the results are shown in Table 1 [25, 38, 39].

$$C = \frac{\int I(V) dV}{(2m \Delta V (V_2 - V_1))} \quad \text{---} \quad (2)$$

where, $\int I(V) dV$ = total voltammetric charge in coulombs, m = mass of the active electrode in grams, ΔV = scan rate in V/s and $(V_2 - V_1)$ = potential window range in V. In Table 1, the specific capacitance values decrease with an increase in the scan rate. This observation is attributed to the fact that at lower scan rates, there is sufficient time for the electrolyte ions to occupy almost all the active sites of the electrode material, resulting

in higher specific capacitance. On the contrary, at higher scan rates, the limited time available for the electrolyte ions to reach the electrode surface leads to a lower specific capacitance.

Chronopotentiometry studies revealed the capacitive nature of the samples. Figures 8(c) & (d) demonstrate the charge-discharge cycles of the samples at various current densities. Nanocrystalline Mn₃O₄ exhibits symmetric charge-discharge curves, indicating high electrochemical reversibility and exceptional capacitive behavior, which corroborates the cyclic voltammetry results. Conversely, the charge-discharge curves of the Co₃O₄ nanocrystals are not symmetric, with lower discharge time. The discharge capacitance (C) was obtained with the help of the formula given below

$$C = \frac{I \Delta t}{\Delta V m} \quad \text{--- (3)}$$

where, I= current in amperes, Δt = discharging time in seconds, ΔV = potential window in volts and m= mass of active electrode material in grams. [25]

Mn₃O₄ and Co₃O₄ nanocrystals demonstrated distinct discharge capacitance values at 0.5 A/g current density, reaching 417 and 214 Fg⁻¹, respectively. These specific capacitance values were calculated and listed in Table 2 for various current densities. The results from the constant potential (CP) studies aligned with the cyclic voltammetry (CV) findings.

A critical factor in supercapacitors is their long-standing cyclic stability. To assess this, Mn₃O₄ charge-discharge cycles were conducted for ten thousand cycles at a current density of 2 A/g (Figure 8(e)). Remarkably, despite 10000 cycles, 81% of capacitive retention was noticed. In contrast, Co₃O₄ exhibited 70% capacitive retention at 0.5 A/g current density (Figure 8(g)). The charge-discharge curves of Mn₃O₄ displayed a symmetric pattern with cycle number, indicating a high level of electrochemical reversibility and excellent capacitive behavior [40-44]. The columbic efficiency was determined with the formula given below (4).

$$\eta = \frac{t_d}{t_c} \times 100 \quad \text{--- (4)}$$

where, η , t_d and t_c are columbic efficiency, discharging time & charge time correspondingly. The Mn₃O₄ electrode showed 100% columbic efficacy beyond six hundred cycles as per Figure 8(e).

Electrochemical Impedance Spectroscopy (EIS) is an alternative characterization method employed for investigating the capacitive behavior of the electrode materials. EIS analysis of the sample is conducted in the frequency range from 1 Hz to 1 MHz to corroborate the CV results. The Nyquist plots of the Mn₃O₄ nanoparticles are shown in Figure 8(f). The juncture of the Nyquist plots on the real axis at greater frequencies signifies the solution resistance (RS). It's witnessed that both the RS and charge transfer resistances of the materials are tinier before cycling and become elevated after 10,000 cycles, that results a decline in the specific capacitance with cycles, like depicted in Figure 8(f) & (h).

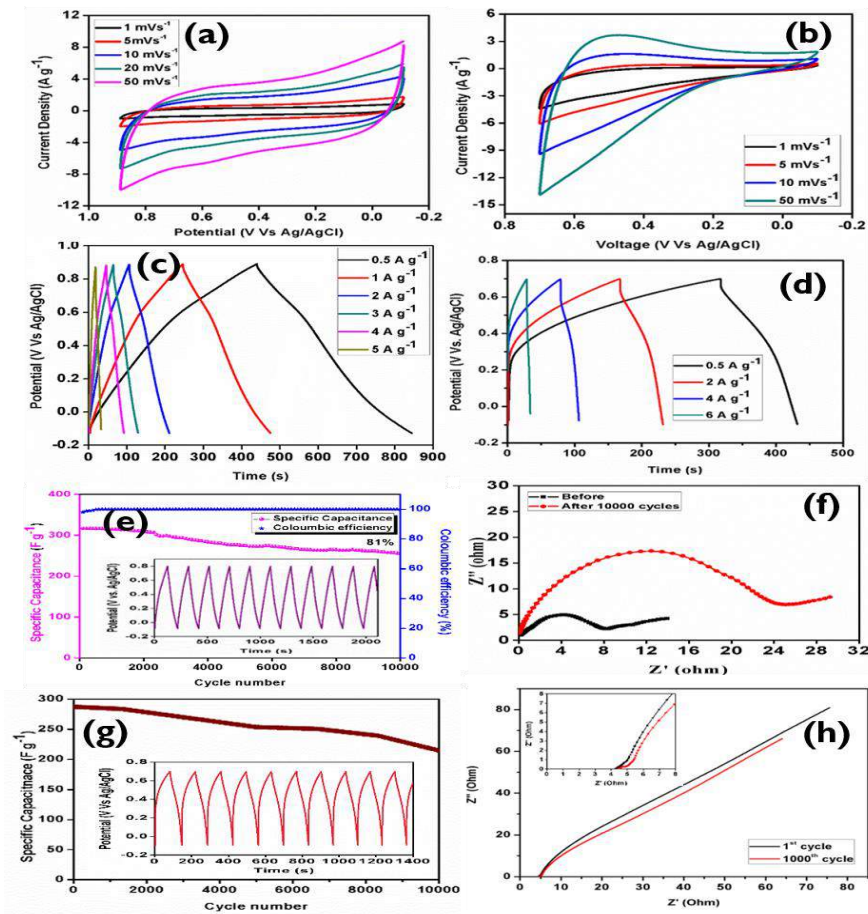


Fig. 8: (a) Spinel Mn_3O_4 - cyclic voltammety curves (b) nanocrystalline Co_3O_4 at various scan rates. (c) Charge-discharge curves of spinel Mn_3O_4 (d) Charge-discharge curves of Co_3O_4 nanocrystals (e) Specific capacitance and coulombic efficacy with cycle number (f) nanocrystalline Mn_3O_4 - Nyquist plots (g) Specific capacitance with cycle number (h) nanocrystalline Co_3O_4 - Nyquist plots

IV. CONCLUSION

Co_3O_4 and Mn_3O_4 , spinel nanostructures were successfully synthesized using the solution combustion technique. XRD and Raman studies confirmed the formation of cubic shaped Co_3O_4 , tetragonal Mn_3O_4 with assessed crystalline sizes of 22 nm and 14 nm, respectively. FTIR results further supported the production of cobalt oxide and manganese oxide, evident from the presence of Co-O and Mn-O bonding. The SEM micrographs showed evenly spread sphere-shaped grains with a typical size of 120 nm for Co_3O_4 and disseminated uneven grains of approximately 100 nm size for Mn_3O_4 . Optical band gaps of the samples were measured to be 2.93 eV for Mn_3O_4 and 2.50 eV for Co_3O_4 nanocrystals. Additionally, they exhibited high ferromagnetism at around 5 Kelvin temperature. The electrochemical characterization studies, including CV, Chronopotentiometry, and EIS, were performed in one Molar sodium sulfate aqueous electrolyte. The as-prepared Mn_3O_4 nanocrystals demonstrated the maximum specific capacitance of 417 F/g at a current density of 0.5 A/g, along with reliable cyclic stability and just about 100% coulombic efficacy. From the above results, we can say that nano crystalline Mn_3O_4 shows promise to be an exceptional material to make electrodes for applications of supercapacitor.

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